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Fatigue Crack Initiation Properties of Rail Steels

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PREFACE

This report presents the results of a program on rail steel fatigue crack initiation characterization. It has been prepared by Battelle's Columbus Laboratories (BCL) under Task 3 of Contract DOT-TSC-1426 for the Transportation Systems Center (TSC) of the Department of Transportation. Work was begun under the technical direction of Roger Steele of TSC and subsequently completed under the supervision of Oscar Orringer of TSC.

The experimental work was performed in the Fatigue Laboratory of Battelle's Columbus Laboratories by Norman Frey, whose care and concern are gratefully acknowledged. The development of nondestructive crack initiation monitoring techniques was handled by Ms. Karen Pfister. Her efforts are also appreciated. Some assistance on data analysis was also gratefully received from Ronald Galliher.

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This report presents part of the results of a study on rail material characterization for the correlation of rail defect growth and failure properties to better define rail defect mechanisms. The work was conducted as part of the Track Structures Research Program under the direction of the Transportation Systems Center and sponsored by the Federal Railroad Administration. The results are presented in two volumes entitled.

Determination of Residual Stresses in Rails

Fatigue Crack Initiation Properties of Rail Steels.

This report describes an experimental study in which the objective was to determine fatigue crack initiation properties of standard American Railway Engineering Association (AREA) rail steel. The fatigue life of structural components is determined by the sum of the loads and stress cycles required to initiate a fatigue crack and to propagate that crack from subcritical dimensions to a critical crack size. Thus, knowledge of fatigue crack initiation properties, which crack propagation models do not address, is necessary for a complete assessment of rail life.

One new and four used rail steels were investigated. The influence on crack initiation behavior of stress ratio, control mode, orientation within the rail and periodic overstrain were investigated. Both constant and variable amplitude experiments were performed. An analytical model was developed, employing an equivalent strain parameter, which allowed prediction of variable amplitude (service load) fatigue crack initiation from constant amplitude material characterization data.

From this study it can be concluded that periodic overstrains above the constant amplitude fatigue limit will substantially increase the damage caused by strain ranges below the limit, and a periodic overstrain fatigue curve should be used in life predictions on such spectra. Accordingly the entire range of traffic loads must be considered, not just isolated overloads. Transverse rail head cracks were found to initiate more rapidly than longitudinal rail head cracks, and the relative susceptibility of various head locations to rail fatigue failure was identified. Also, it was demonstrated that linear damage accumulation models can be used effectively to provide reasonable crack initiation life predictions.

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A typical failure in a rail is the culmination of a progressive damage process that begins with the initiation of a fatigue crack, followed by the growth of that crack to a critical size. If the reliability of a rail system is to be assessed, it is quite important to develop data on the crack initiation and propagation behavior of rail materials and to develop a damage accumulation model which relates constant amplitude laboratory fatigue data with variable amplitude and service simulation fatigue behavior.

Crack propagation data, both constant and variable amplitude, were developed in a previous DOT/TSC program (1)* (Contract No. DOT-TSC-1076). A predictive model for crack growth was also developed. (2)

In task 3 of this program, the problems of crack initiation were assessed, both in terms of the development of critical data and the formulation of a damage model. This report contains the results of the Task 3 efforts.

References are listed on page 56.

2. RAIL MATERIALS

Two 39-foot sections of new 100-pound/yard rail steel were purchased from the Fritz Rumer Cooke Company for this program. In addition, four shorter sections of used rail material were obtained to provide a cross-section of used rails for a comparison with presently produced standard rail. Background data on the used rail materials are presented in Table 1. Further information on chemistry is included in Reference 3 and an inclusion content in Reference 4.

BCL Rail Number	Source Number	Rail Weight (1bs)	Year Rolled	Characteristics
3	100	85	1920	Low Sulfur/Oxygen Ratio High Inclusion
4	418	130	1929	Low S/O Ratio Low Inclusion
5	VD-2	115	1974	High S/O Ratio Intermediate Inclusion
6	398	130	1929	Intermediate S/O Ratio Intermediate Inclusion

TABLE 1. USED RAIL MATERIAL BACKGROUND DATA

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3. EXPERIMENTAL DETAILS

3.1 SPECIMENS

Two specimen geometries were used for the majority of experiments in this program. Both specimen geometries were designed with a 0.25-inch diameter reduced section. The specimens tested at BCL were designed with threaded ends as shown in Figure 1, while the specimens tested at Boeing Commercial Airplane Company (BCAC) were designed with longer smooth ends as shown in Figure 2, to accommodate hydraulic grips. A small number of reduced-size, threaded-end specimens of the type shown in Figure 3 were also designed for transverse rail orientation specimens.

The longitudinal rail specimens were removed from three different locations within the rail cross-section. The center head specimens were taken from 3/4-inch square blanks of material centered on the midplane on the rail head (see Figure 4). The intermediate head specimens were taken from blanks centered 1/2-inch from the rail midplane and one-inch from the upper rail surface (see Figure 5). The surface head specimens were taken from blanks taken as near the side surfaces of the rail as possible centered in a plane one-inch from the upper rail surface (see Figure 6). The transverse rail head specimen blanks were taken from the rail centered in a plane oneinch from the upper rail surface. Because of the location and orientation within the rail, the transverse specimens were only about 2-1/2 inches long, as previously shown in Figure 3.

The number of each type of specimen produced is detailed in Table 2. An overall total of 254 specimens were produced; 156 of those for testing by BCAC, including 64 used rail specimens.

Rail	Specimen Type	BCL Specimens	BCAC Specimens
Unused	Center	78	42
	Intermediate		20
	Surface	12	30
	Transverse	8	
Used	Center		_ <u>64</u> ^(a)
	Total	98	156
	OVERALL TOTAL	· · · · · · · · · · · · · · · · · · ·	254

TABLE 2. RAIL SPECIMEN INVENTORY

(a) 16 specimens from each of 4 used rails.

3.2 TESTING PROCEDURES

The BCL fatigue crack initiation experiments were conducted in a 20,000-pound capacity electrohydraulic servocontrolled fatigue machine. All BCL tests were performed in strain control at cycling frequencies ranging from 5 to 20 Hz, depending on strain amplitude. Environmental conditions were maintained at 70F and 50-percent relative humidity. Load as a function of time was recorded continuously for most experiments and stress-strain hysteresis loops were recorded at frequent intervals.

The BCAC crack initiation experiments were conducted on an Amsler Vibraphone in load control at a frequency of about 105 Hz. In selected cases, the specimen temperature during testing was monitored; temperatures approaching 150 F were found in short





FIGURE 2. BCAC UNIAXIAL FATIGUE TEST SPECIMEN



FIGURE 3. TRANSVERSE RAIL UNIAXIAL FATIGUE TEST SPECIMEN



FIGURE 4. CENTER HEAD SPECIMEN LOCATION



FIGURE 5. INTERMEDIATE SPECIMEN LOCATION



life tests (N_i \approx 100,000 cycles) while temperatures did not exceed 100 F for the long life tests (N_i \geq 1,000,000 cycles).

3.3 CRACK INITIATION DETECTION

It was a primary objective in the BCL tests performed in this program to develop some technique for detection of small fatigue cracks in the laboratory specimens tested under constant amplitude conditions. Detection of cracks less than 0.04 inch (1 mm) deep was considered possible and desirable. This would provide a less ambiguous definition of crack initiation, necessary for a smooth transition from crack initiation analysis to crack propagation analysis in the reliability study.

Originally, it was considered likely that either ultrasonics or eddy-currents could be used to detect these small fatigue cracks. Ultrasonics in the pulse echo mode was at first thought to be the best approach because it was easily adaptable to automated testing. One of the anticipated difficulties with pulse-echo ultrasonics, however, was that the reflection from the interface might obscure the reflection from the crack. By using a frequency above 10 MHz, it was thought the beam spread would be negligible and the wave length small enough to detect the crack. Although a higher frequency means a more directive beam according to the formula for beam spread, sin $\theta/2 = 1.22\lambda$, there is a tradeoff involved because the higher frequency pulses are more easily scattered by the material, resulting in loss of resolution.

1 -.

The adequacy of the ultrasonics approach was evaluated experimentally in the laboratory using a 0.25-inch-diameter Aerotech (Alpha) transducer at frequencies of 5, 10, 25, and 43 MHz. The ultrasonic pulse generators were the Sperry 721 Reflectoscope, Sonic Mark I, and the Matec Pulse Modulator and Receiver Model 6600. A Lucite holder grip was machined to support the transducers longitudinal to the axis of the specimen and maintain a specific free gap between the end of the specimen and the transducer crystal, as in Figure 7. In the gap between transducer and specimen was an oil couplet basin acting as a cushion for the crystal face during fatigue cycling.

The results of the ultrasonic pulse-echo testing proved to be negative at all frequencies and with all instruments. It was concluded that the filleted end of the gauge section caused mode conversions (change from longitudinal waves to shear waves) and extensive back reflections of such high amplitide that the signal to noise ratio was very low and that any reflection from the EDM notch was obscured, and virtually hidden in the noise (Figure 7). The high frequencies used also caused scattering and loss of resolution. This result was disappointing in view of previous positive results obtained by NASA Lewis. (5) This earlier work was done on an unfilleted, notched specimen, however, and the negative effect of the fillet on the signal to noise ratio was substantial.

Since ultrasonics proved to lack the sensitivity required, eddy-current detection was evaluated. An NDT 15 Eddyscope was used with two different Nortec surface frequency probes (500 KHz and 2 MHz models). These probes were 1/4-inch in diameter, enclosed within a casing and spring loaded for uniform contact on the specimen.

The NDT 15 Eddyscope in combination with the 500 KHz surface probe proved successful in the laboratory for detection of fatigue cracks as shallow as 0.015 inch. Its major drawback was that it could not be used as a continuous monitor of crack initiation. Every time a check was made with the eddy current system, the machine had to be shut down and the strain control extensometry had to be removed from the specimen. This proved to be very time-consuming and inefficient.

Because of problems with both the ultrasonic and eddy current techniques, a third crack detection method was investigated. This method was based purely on the detection of a change in specimen compliance that occurs when a crack forms. One significant problem was found in differentiating between an apparent compliance change caused by cyclic hardening or softening of the material and an actual compliance change caused by crack formation. It was found, however, through some trial and error experimentation, that nearly all specimens developed a stable stress response to constant amplitude strain cycling by the time each specimen has been subjected to about half the expected cycles to failure. At this point, it was possible to monitor the maximum cyclic stress (actually load was measured but for a constant specimen area on an unnotched specimen, stress can be considered directly proportional to load) and to note small increases or decreases in that maximum stress. If such changes occurred, it was nearly always an indicator that a crack had developed and that failure of the



FIGURE 7. ULTRASONIC TRANSDUCER SETUP



FIGURE 8. HEAT TINTED CRACK INITIATION ZONE ON SPECIMEN 2-11-E

specimen was soon to follow. A decrease in max. stress indicated cracking in the specimen between the clip gauge probes because the compliance of a specimen reduces in the region of a crack which lowers the load for a given strain limit. An increase in max. stress indicated a crack outside the probes of the clip gauge. The large majority of specimens developed cracks between the clip gauge probes.

The specimen compliance method of crack detection was verified on a series of specimens at different strain ranges. The limit detectors on maximum stress were set to shut off the machine when the maximum stress (after stabilization) varied by 2 to 4 percent. After the machine shut down, the specimen was removed and placed in a small furnace at about 700 F for 45 minutes. This heating process caused a tinting of the crack which was visible on the fracture surface after the specimen was reinstalled in the test system and cycled to failure. Figure 8 shows the heat tinted crack initiation on one of the test specimens cycled at a strain range of 0.80 percent.

Through the above procedures, it was possible to record cycles to initiation, the crack depth at initiation, and cycles to failure for a number of the constant strain amplitude tests performed at BCL. There was considerable scatter in the final results (see Figure 9) but the expected trend was evident — an increasing ratio of initiation cycles to propagation cycles (N_i/N_f) for increasing size of the initiated crack. Crack depths below 0.010-inch were not plotted because of measurement uncertainties. Observed scatter is the result of several factors — (1) multiple cracks in some instances (2) strain control cycling which allows cracks to propagate in a semistable manner (especially under low strain amplitudes), and (3) inherent scatter in material behavior. In the extreme, if the load drops enough with a ductile material, it is possible for the crack to act essentially as an elastic hinge which results in unrealistically high N_i/N_f ratios. For these reasons, the experimental results did not permit establishment of a clearly defined relationship between N_i/N_f

In summary, the specimen compliance technique could not be used to precisely identify the point when a small crack of some specific depth has been reached. Too many variables influenced the accuracy of the technique. It could be used, however, to identify an approximate relationship between N_i/N_f and crack depth that could be used to approximate what portion of the total cycles to failure was involved in the initiation of a crack of a given depth. It also was a method of crack initiation monitoring that could be used both by BCL and BCAC. It was, therefore, adopted by both laboratories in this study.

4.1 CONSTANT AMPLITUDE EXPERIMENTS

A total of 191 constant amplitude crack initiation experiments were completed in this program. Of this total, 42 were strain-controlled experiments conducted at BCL, while the others were load-controlled experiments conducted at BCAC. All of the BCL experiments were completed on new rail material, while 57 of the BCAC specimens were taken from used rail material. The program involved extensive laboratory testing approximately 200 million fatigue cycles were applied to generate just the unused rail data.

4.1.1 Unused Rail Data

A direct comparison of data generated in both laboratories was desirable, but that comparison was complicated by the differences in control parameters. In the BCL strain control tests, cyclic softening of the new rail material from its initial monotonic stress-strain response was observed. This behavior is shown in Figures 10, 11, and 12 for center, surface, and transverse rail specimens, respectively. In other words, for a given strain amplitude, the observed maximum stress in individual hysteresis loops decreased from a higher initial monotonic response to a lower, stable maximum stress. A corresponding increase in plastic strain was, of course, seen with the decreasing maximum stress in each specimen. This trend is clearly visible in a series of hysteresis loops reproduced in Figure 13.

In the BCAC load control tests, strain was not monitored, but it is evident from Figures 10 and 11 that some cyclic softening (under constant stress) should be expected at the stress levels evaluated. With small plastic strains, as were seen in the long-life BCAC tests, this softening behavior remained stable and controlled; but load control tests at higher stress amplitudes would likely have resulted in unconstrained cyclic strain softening (or ratcheting) of the specimen, leading to tensile failures rather than fatigue failures. Since only medium to long life tests were performed at BCAC, this ratcheting phenomenon did not occur.

From the cyclic stress-strain data developed in the BCL tests, it was possible to predict the stable strain response of the BCAC tests and to equate test results from the two sources. This was done through calculation of an equivalent strain parameter previously analyzed at BCL. (6) This parameter is similar in form to that originally developed by Walker, (7) but it was modified along the lines suggested by Smith, et al. (8) to define ε_{eq} as follows:

$$eq = (\Delta \epsilon)^m (\sigma_{max})^{1-m}$$

where $\Delta \varepsilon$ = stable strain range

 σ_{max} = maximum stable stress

E = elastic modulus

m = constant for a material

In order to compute equivalent strains using Equation (1), it is obviously necessary to know the correct values for each constituent term in the expression. In the BCL strain control tests $\Delta \epsilon$ was controlled, σ_{max} was measured (at the point of stable stress response), and an average E was computed from a series of monotonic stress-strain curves. In the BCAC load control tests σ_{max} was controlled, but $\Delta \epsilon$ and E had to be estimated from monotonic and cyclic stress-strain data. Values of $\Delta \epsilon$ were computed by using the following equations which approximated the cyclically stable stress-strain response of the longitudinally oriented rail specimens:

$$\Delta \varepsilon = \frac{2S_a}{E} , \quad 0 \le \Delta \varepsilon \le \Delta \varepsilon (1) , \quad (2a)$$

$$\Delta \varepsilon = 2 \left(\frac{S_a}{K_1} \right)^{1/n_1}, \quad \Delta \varepsilon (1) < \Delta \varepsilon \leq \Delta \varepsilon (2), \quad (2b)$$

(1)



. . . :









MONOTONIC AND CYCLIC STRESS-STRAIN BEHAVIOR OF CENTER-HEAD RAIL SPECIMENS FIGURE 10.

A.



FIGURE 11. MONOTONIC AND CYCLIC STRESS-STRAIN BEHAVIOR OF SURFACE-HEAD RAIL SPECIMENS









 $\Delta \varepsilon = 2 \left(\frac{S_a}{K_2}\right)^{1/n_2}, \ \Delta \varepsilon (2) < \Delta \varepsilon \quad , \qquad (2c)$

where

 K_1 and K_2 = strength coefficients

 n_1 and n_2 = strain hardening exponents

 $\Delta \varepsilon(1)$ and $\Delta \varepsilon(2)$ = specific values of $\Delta \varepsilon$ that denote break points in the tri-log-linear stress-strain curve approximation.

The approximate values for the constants in the Equation (2) series were found as follows for the new rail material:

 $E = 29 \times 10^{3} \text{ ksi}$ $K_{1} = 3.43 \times 10^{3} \text{ ksi}$ $\Delta \varepsilon (1) = 2.7 \times 10^{-3}$ $n_{1} = 0.677$ $R_{2} = 8.05 \times 10^{2} \text{ ksi}$ $\Delta \varepsilon (2) = 4.0 \times 10^{-3}$ $n_{2} = 0.443$ $S_{a}(1) = 39.1 \text{ ksi}$ $S_{a}(2) = 51.3 \text{ ksi}$

A value for the material constant, m, in Equation (1) was found through an examination of the BCAC data generated at three stress ratios. By comparing the stress conditions which provided nearly identical fatigue lives at different stress ratios, it was possible to iteratively solve Equation (1) to find a value of m which provided equal equivalent strain values for conditions where equal fatigue performance had been found. A value of m = 0.6 gave the best overall consolidation of the data.

All of the crack initiation test data developed on unused rail samples in this program are listed in Table 3. The data include specimen type, source, stress ratio, maximum stress, computed or measured total strain, computed equivalent strain and specimen identification. The fatigue lives are also listed in increasing order of cycles to failure for each condition, and appropriate group statistics are presented including average fatigue life (based on the antilog of the average of the log lives), and the coefficient of variation. The coefficient of variation is defined as the ratio of sample standard deviation to the mean (in percent); it provides an indication of the relative variability within a data set. It should be noted that the cycles to initiation for the BCL tests were adjusted to 94 percent of total cycles to failure because that ratio of N_i to N_f compared approximately with a crack initiation crack depth of 0.030 to 0.040 inch, which is the approximate crack initiation crack size chosen for the reliability analysis. The number of non-runouts and total specimens tested in each category are listed in the last two columns of Table 3.

The fully reversed (R = -1.0) results presented in Table 3 are displayed graphically in Figures 14 and 15 for center head specimens and other specimen locations, respectively. Good agreement between BCL and BCAC test results is evident. There is also close agreement between fatigue data from center head specimens and surface and intermediate specimens; data for transverse specimens fall well below those for other orientations. Both the longitudinal and transverse orientation crack initiation fatigue test results compare favorably with data generated in an earlier program at BCL (9) on a hot rolled rail material. Substantial differences are evident between the data developed in this program and those developed by Fowler, (10) however. For example, his indicated endurance limit stresses (R = -1.0) range from about 52 to 62 ksi for six different rail materials. The BCL and BCAC results indicated endurance limit stresses below 50 ksi. It is likely that Fowler data fall higher because of the rotating beam test equipment which he used. Rotating beam tests inherently apply maximum stresses to a small volume of material at the surface of the material, which commonly leads to the infamous size effect in fatigue. (11)

4.1.2 Used Rail Data

The crack initiation tests on the used rail materials were conducted at BCAC. Including specimens used to set stress levels, there were 57 tests completed on four different used rails. The tests were performed in load control at a stress ratio of 0.10 with maximum stress levels ranging from 65 ksi to 105 ksi. Five to seven tests TABLE 3. COMPLETE LISTING OF ALL BCL AND BCAC CONSTANT AMPLITUDE CRACK INITIATION TEST DATA ON UNUSED RAILS

Tests	Total	2	-	'n	, م	-	~	∞
Number of	Non- Runout	2	F	ະ ຕ	ъ	-	4	2
Coefficient of	Variation, %	0.7	•	2.1	5.7	ı	5.7	6.3
Average Life Based	on Log, n _i	11,260	14,330	60,595	103,982	104,000	223,615	601,446
Ranked Fatigue	Life n _i	10775 11770	14330	49710 57610 77690	60000 65000 88000 115000 308000	104000	92850 163800 183300 187800 234700 246800 224600	256000 328000 431000 438000 610000 973000 3026000 5087000R
Specimen	Identi- fication	1-6-Е 2-11-Е	2-3-C	2-2-C 1-12-C 2-12-E	2-11-D 1-6-D 2-10-B 2-13-B 2-5-B	1-7-B	1-2-A 1-11-C 1-10-A 1-2-E 1-5-E 1-5-A 2-3-E 2-3-E	1-10-8 2-11-8 2-24-D 1-3-D 1-3-D 1-7-D 2-3-8 1-7-D 2-8-D
Computed Equivalent	Eeq x 10-3	4.86	4.49	3.42	3.15	2.95	2.89	2.80
Computer or Measured Total	Strain, ∆e%	0.80	0.70	0.50	• 0.448	0.412	0.40	0.388
Maximum	Stress, ^o max	66.8	67.0	56.3	54.0	52.0	51.4	20.0
Stress	katio, R	-1.00	•		<u> </u>			· · ·
	Source	BCL	BCL	BCL	BCAC	BCAC	BCL	BCAC
	Jype Type	Center						

					_							·								-	-	
	F Tests	Total	13		_					_	-	د		-	-	-	4			•	m	
-	Number of	Non- Runout	9	<u>.</u>							0	0		0	0	0	4		r	-	ĉ	
	Coefficient of	Variation, %	5.6								-			-	1	 	2.3		-		1.6	,
	Average Life Based	on Log, n _i	657,367	-										1	-	 	116,784		334.000		158,577	-
	Ranked Fatique	Life n ₁	286300 322700	398700	1241000	1401000	5000000R 5000000R	500000R	5870000R	810000R	5078000R	1680000R	1420000R	5187000R	5870000R	7250000R	91000 000201	120000	334000		127870 173940	179290
	Specimen	Identi- fication	2-9-C 1-1-C	л-1- 1-1-	- 1-5-C	1-1-A	2-7-C	2-9-E	2-5-A 2-7-A 2-4-A	1-2-C	2-4-B	1-10-E	2-2-7 1-8-É	2-7-D	,1-4-E	2-12-C	1-6-D	1-10-B	2-7-B	3 . 1	2-8-C 2-10-F	2-10-C
100) · · · · · · · ·	Computed Equivalent	Strain ^e eq x 10 ⁻³	2.73								2.73	2.53		2.52	2.42	2.25	3.15		2 QF		2.89	
	Computer or Measured Total	Strain, ∆ε%	0.375					- ,			0.376	0.35		0.343	0.33	0.30	0.448	*	0.412		0.40	-
	Maximum	Stress, o max	49.2								49.0	45.3		46.0	44.2	42.5	54.0		52 . D		51.6	
	Stress	Ratio, R												· .			-1.00			-	,	
		Source	BCL				•	· ·	<u>.</u> .		BCAC	BCL		BCAC	BCL	BCL	BCAC		BCAC	2.5.	BCL	
·		Specimen Type							•			,		·			Surface					

TABLE 3. (Continued)

TABLE 3. (Continued)

- Tests Total	Q	ε Γ	- 2	4	F N	••••••••••••••••••••••••••••••••••••••	ی – م		-
Number of Non- Runout	4	8	0 0	4	- 2	m	- S	- - -	` -
Coefficient of Variation,	1.	6. <u>6</u>		3.2	 2.6	4.0	3.3		-
Average Life Based on Log, n _i	393,044	937,365	 91,940	901,653	2816,000. 44,725	57,526	57,000 122,619		294,000
Ranked Fatigue Life n _i	303000 309000 358000 712000 5080000R 5100000R	528100 1663800 1520000R	5116000R 79000 107000	589000 645000 1314000 1324000	2816000 36848	54285 42770 46568 95579	57000 66000	107000 144000 154000 177000	294000
Specimen Identi- fication	1-8-E 1-10-D 2-8-D 2-8-D 2-11-D 1-5-D 2-13-B 2-13-B	2-8-A 2-10-A 2-7-D	1-5-8 2-12-D1 2-13-D1	1-13-82 1-13-81 1-13-81 1-13-02 2-12-81	2-12-B2	1-1-T 2-4-T 1-2-T 2-3-T	2-6-B 1-3-B	2-9-D 1-11-D 2-1-D 2-6-D	1-2-8
Computed Equivalent Strain Eeq x 10-3	2.80	2.73	2.66 3.15	2.80	2.73 2.86	2.72	3.37 3.15		2.94
Computer or Measured Total Strain, ∆∈%	0.388	0.375	0.365	0.388	0.376 0.40	0.375	0.332 0.308	·	0.284
Maximum Stress ^G max	50.0	49.1	48.U 54.0	50.0	49.0 50.1	48.7	100.0 95.0		90.06
Stress Ratio, R		· · · · · · · · · · · · · · · · · · ·	-1.00	· <u> </u>	-1.00		0.10		
Source	BCAC	BCL	BCAC	BCAC	BCAC BCL	BCL	BCAC BCAC		BCAC
Specimen Type			Inter- mediate		Trans-	Verse	Center		

Number of Tests	Non- Runout Total	6		-	0	-	4 4	4	- 0	2	4 4	- - - - -
 Coefficient of	Variation, %	4.1					1.2	5.4	;	2.2	1.11	
Average Life Based	on Log,	484,608		3384,000		62,000	122,087	839,102		60,597	469,715	<u>.</u>
Ranked Fatique	Life n	220000 274000	495000 651000 771000 806000	3384000	5068000R	62000	108000 117000 118000 149000	423000 589000 861000 2311000	5026000R	51000 72000	112000	2458000
Specimen	Identi- fication	1-4-D 1-4-B	2-3-0 2-1-8 2-7-8 2-10-0	2-9-B	1-10-D	1-9-D	2-6-D 1-8-D 2-11-8 2-8-8	1-11-D 1-3-B 2-3-D 1-9-B	2-9-D	1-1-81 2-13-02	2-12-D2 1-10-1 1-1-82	2-2-D]
Computed Fouivalent	Strain Eeq x 10-3	2.75		2.69	2.23	3.37	3.15	2.75	2.43	3.15	2.75	
Computer or Measured Total	Strain, $\Delta \epsilon \%$	0.264		0.258	0.214	0,332	0.308	0.264	0.233	0.308	0.264	
Maximum	Stress, ^o max	85.0		83.0	69.0	100.0	95.0	85.0	75.0	95.0	85.0	
Stress	Ratio, R			<u>-</u>	-	0.10		• 4	-	0.10		
``````````````````````````````````````	Source	BCAC		BCAC	BCAC	BCAC	BCAC	BCAC	BCAC	BCAC	BCAC	
	Specimen Type					Surface	<u> </u>	· · · · · · · · · · · · · · · · · · ·		Inter- mediate		

TABLE 3. (Continued)

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		· ·						
. Tests	Total	9		4	4	2	m	-
Number of	Non- Runout	و		4	4	2	2	0
Coefficient of	Variation, %	7.0		4.7	6.8	1.4	3.1	
Average Life Based	on Log, n _i	1082,621		164 ,546	739,432	287 ,757	1342,642	1
Ranked Fatique	Life n:	338000 685000 1716000 2371000	3800000 4498000	81000 157000 183000 315000	355000 391000 839000 2567000	254000 326000	984000 1832000 5059000R	5058000R
Specimen	Identi- fication	1-5-D 1-8-8 1-8-0	1-5-8 2-8-8	2-10-8 2-6-8 2-7-0 1-7-0	1-3-D 2-10-D 1-7-B 2-9-B	1-1-D2 2-2-B1	1-13-01 1-11-82 2-2-02	2-2-82
Computed Equivalent	Strain E _{eq} x 10-3	2.62		2.84	2.62	2.84	2.62	2.50
Computer or Measured Total	Strain, ∆ɛ%	0.198		0.216	0.198	0.216	0.198	0.190
Maximum	Stress, ⁰ max	115.0		125.0	115.0	125.0	115.0	110.0
Stress	Ratio, R			0.50		0.50		
- -	Source	BCAC		BCAC	BCAC	BCAC	BCAC	BCAC
	Specimen Type		,	Surface		Inter- mediate		· ·

TABLE 3. (Continued)







FIGURE 15. FATIGUE LIFE VS. EQUIVALENT STRAIN FOR SURFACE-, INTERMEDIATE-, AND TRANSVERSE-HEAD RAIL SPECIMENS, R=1.0

were completed at each of two stress levels for each rail material. The reduced data are presented in Table 4. The variability in fatigue performance compared to the baseline data on unused rail material is shown in Figure 16. The data are plotted in terms of maximum stress rather than equivalent strain since cyclic stress-strain data on the used rail materials were not generated.

Two used rail materials performed far below crack initiation fatigue life trends previously established for the unused rail. These were rails produced in the 1920's. One rail performed similarly to the new, unused rail; it was a vacuum-degassed rail produced since 1970. The fourth rail, which performed somewhat better than the unused rail, was also produced in the 1920's. No correlation was evident between fatigue performance and either sulfur to oxygen ratio or inclusion content of the rail material.

#### 4.2 PERIODIC OVERSTRAIN EXPERIMENTS

The objective of these experiments was to determine the extent to which periodic overstrains would affect the fatigue resistance of the rail steel in the regime of the constant amplitude fatigue limit and below. If it were assumed that all cycles experienced by a rail material below its constant amplitude fatigue limit are nondamaging, a periodic overstrain should not cause failure until the total number of overstrain cycles is equal to the cyclic life of a virgin specimen subjected to constant amplitude cycling at the overstrain level. A comparison of actual cycles to failure and this hypothetical fatigue life is made in Table 5. It is obvious that the small amplitude cycles below the constant amplitude endurance limit were actually quite damaging when combined with periodic overstrains. In fact, if the small amplitude cycles to failure are plotted on an equivalent strain basis, excluding the overload cycle, the endurance limit essentially vanishes (or is reduced appreciably), This is shown in Figure 17 where the overstrain data are plotted relative to the constant amplitude baseline crack initiation curve. Elimination of the constant amplitude endurance limit through periodic overstrains has been observed previously by Brose.(13)

Figure 17 does not clearly illustrate the relative damage caused by the overstrain and small amplitude cycles. Table 6 was constructed in an effort to evaluate the extent of damage caused by the large and small amplitude strain cycles; cycles to failure are noted, along with the number of overstrain cycles endured. The number of overstrain cycles were divided by the average cycles to failure under a constant strain range of 0.80 percent (11980 cycles), to compute a percent of damage caused by the large cycles (assuming the linear damage theory as developed by Miner is valid). (12) The number of small amplitude cycles are also listed. Theoretically, according to the linear damage hypothesis, the balance of damage not caused by the large overstrain cycles (100 percent = total damage of failure) should have been caused by the small strain cycles. If this were true, the damage per small cycle could be computed as follows:

$$\Delta D_{small} = 100 - \Delta D_{large}, \ \ , \qquad (3)$$

where  $\Delta D_{small}$  = percent of damage caused by small cycles

 $\Delta D_{large}$  = percent of damage caused by large cycles

N_{small} = number of small cycles.

Values for Equation (4) are presented in the last column of Table 6 for each overstrain specimen.

If the damage process were indeed linear, it would have been expected that the damage/cycle for all four tests at the two different small strain cycle levels would have been equal, or nearly so (taking into account normal fatigue data variability). Differences do exist, however, which suggest some history dependence within the damage process. The history dependence is most evident for the very small strain range tests (0.3 percent). There are two readily identifiable explanations of this phenomenon, neither of which appear to be completely satisfactory, or readily applied to a complex stress history.

	COEFFICIENT OF VARIATION, CYCLES	1.3 3.4 ^a
	6 (398) AVERAGE FATIGUE LIFE CYCLES	85,110 354,800 ^a
	MAXIMUM STRESS, ksi	105 95
er	COEFFICIENT OF VARIATION, CYCLES	4.2 8.1
(ESULTS Source Num	5 (VD-2) AVEŘAGE FATIGUE LIFE, CYCLES	102,300 478,600
JE TEST R umber and	MAXIMUM STRESS, ksi	95 85
RAIL FATIGU BCL Rail N	COEFFICIENT OF VARIATION.	2.3 12.5
OF USED	4 (418) AVERAGE FATIGUE LIFE, CYCLES	56,230 489,800
SUMMARY	MAXIMUM STRESS, ksi	85 65
rable 4.	COEFFICIENT OF VARIATION, CYCLES	2.2
- · · · · · · · · · · · · · · · · · · ·	3 (100) AVERAGE FATIGUE LIFE, CYCLES	128,800 616,600
	MAXIMUM STRESS, ksi	85 75

(a) One very low test value excluded with N  $_{\rm f}$  = 10,000 cycles.

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Stress Level 1  $N_1 = \frac{10^5}{10^5}$ (5 tes ts/rail).

Stress Level 2. N. - 106 (7 tests/rail)



Constant amplitude)

0.300 %

105

-2.60

-2.65 L

strain level

FIGURE 17.

Cycles to Fallure, N₁

PERIODIC OVERSTRAIN FATIGUE TEST RESULTS

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TABLE 5. PERIODIC OVERSTRAIN TEST RESULTS ON NEW RAIL MATERIAL

SPECIMEN IDENTIFICATION 2-5-C 2-10-E 2-1-A 2-12-A 2-10-C 2-8-E 2-3-A 2-7-E CYCLES TO FAILURE EXPECTED IF NO DAMAGE ASSUMED FROM 1,17 11.,250,000 1,125,000 11,250,000 270,200 289,400 900,700 930,785 422,820 401,270 209,240 217,600 STABLE STRESS RANGE, ksi 98.0 93.5 79.2 76.2 74.9 93.8 91.2  $\begin{array}{c} 0.032 ^{(a)} \\ 0.029 ^{(a)} \end{array}$ PLASTIC 0.011 0.037 0.030 STRAIN RANGE, PERCENT ELASTIC 0.345 0.338 0.289 0.296 0.268 0.271 FREQUENCY OF OVERSTRAIN 10 every 10,000 10 every 10,000 10 every 1,000 10 every 1,000 OVERSTRAIN CYCLE 0.8000.800 0.8000.800 STRAIN RANGE, PERCENT SMALL 0.375 0.375 0.300 0.300 CONDITION 2

(a)plastic strain values measured shortly after overstrain cycles were applied.

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CONDITION	CYCLES TO FAILURE, ^N f	OVERSTRAIN CYCLES, ^{AD} Large	OVERSTRAIN CYCLES(a) ^{ΔD} Large	AMPLITUDE CYCLES, N Small	COMPUTED INCREMENT OF DAMAGE PER SMALL CYCLE(D). x 10-4, % ^{AD} Small Cycle
1	209,240	2,090	17.4%	207,140	3.99
	217,600	2,170	18.1%	215,430	3.80
2	270,200	270	2.25%	269,930	3.62
	289,400	289	2.41%	289,111	3.38
3	422,820	4,220	35.2%	418,600	1.55
	401,270	4,010	33.5%	397,260	1.67
4	900,700	900	7.51%	899,800	1.03
	930,785	930	7.76%	929,850	0.99

## TABLE 6. COMPUTATION OF RELATIVE DAMAGE CAUSED BY LARGE AND SMALL CYCLES IN PERIODIC OVERSTRAIN

(a)  $N_f = 11,980$  at  $\Delta \epsilon = .80\%$ 

(b) 100% damage equals failure.

The first explanation is based on a plastic strain accumulation damage hypothesis as investigated by a number of researchers. (14,15) Along this vein, it could be hypothesized that a certain number of large strain cycles were required to cyclically soften the material to the point where plastic strain damage began to develop at the lower strain range as well. Some simple calculations suggest that this is at least plausible, as is demonstrated in Table 7. Since the material does undergo a gradual softening when subjected to cyclic plastic deformation, it is not unreasonable to assume that it would take longer to soften to a level where significant plasticity was experienced at 0.3 percent strain range, than at 0.375 percent. Table 7 suggests that this difference in the required number of large cycles could have been as great as a factor of ten, from 40 large cycles to 400.

The second explanation for the apparent history effects in the overstrain experiments is based on an initiation-propagation damage concept. With this concept, it can be assumed that only cycles above the constant amplitude endurance limit contribute to the formation and initiation of a fatigue crack; but all cycles, including the small ones, are effective in propagating that crack beyond some "equivalent initial flaw." If this concept has physical meaning, it suggests that very small flaws are subject to propagation by sub-endurance limit strain ranges. For example, if it is assumed that the small cycles listed in Column 4 of Table 7 effectively represent initiation cycles, it is evident that  $N_i/N_f$  ratios would be extremely small for a .375 percent strain range (from 1 to 11 percent). If Figure 9 is extrapolated back to such small  $N_i/N_f$  values, it suggests initiation crack depths well below 0.001 inch. At present there is very little experimental evidence to support this concept; therefore, it is not a practical method to account for stress history effects.

#### 4.3 VARIABLE STRAIN AMPLITUDE EXPERIMENTS

The objective of these experiments was to develop a collection of variable amplitude fatigue data for simulated rail histories that could be used to assess current linear damage accumulation prediction capabilities.

Three different spectra were used in the variable amplitude experiments. These spectra were derived from cumulative probability curves of wheel rail loads measured for four different railroads. These curves are shown in Figure 18. The most severe COMPARISON IN NUMBER OF NONDAMAGING SMALL CYCLES IN PERIODIC OVERLOAD EXPERIMENTS TABLE 7.

ι.	AVERAGE DAMAGE PER SMALL CYCLE x 10-4, %	3.91	3.91	1.78	1.76	
	AVERAGE NUMBER OF DAMAGING SMALL CYCLES	210,420	249,800	368,930	524,830	
	NUMBER OF SMALL CYCLES PAST DURING SOFTENING PROCESS	3,000	30,000	39,000	390,000	
	NUMBER OF LARGE CYCLES REQUIRED TO SOFTEN MATERIAL TO LEVEL WHERE PLASTIC STRAIN OCCURS AT LOWER STRAIN LEVEL	40	40	400	400	
	LOWER STRAIN LOAD, %	0.375	0.375	0.300	0.300	
	CONDITION	-	2	e e	4.	

* A block of ten large overstrain cycles began each test, therefore small cycles endured after x overstrain • (R), where R is the ratio of small to large cycles. (ol-x) 0 cycles = -



FIGURE 18. CUMULATIVE PROBABILITY PLOT OF PEAK VERTICAL WHEEL RAIL LOADS

spectrum used in this program was based on the NEC wheel rail loads, while the least severe was based on the SP wheel rail loads. The third spectrum used in this program was based on a combination of the UP and FEC wheel rail load distributions; it fell intermediate to the NEC and SP spectra.

Each of the three basic spectra was converted from cumulative probability curves to load exceedance diagrams for 1-million gross tons (MGT) of traffic. For estimating purposes, 3700 axle passes (peak load occurrences) per day were considered to represent an annual traffic load of about 20 MGT. This meant that  $365 \times 3700/20 = 67,000$  axle passes represented 1 MGT of traffic. In the earlier crack growth work done by BCL on rail steel, (1 § 2) the wheel rail load data were converted to rail stress spectra which were subsequently converted to load histories used to control the fatigue machine during the crack growth experiments. In this program, the previously computed stress ranges were converted linearly to strain ranges. All strain ranges in each spectrum were programmed to be proportional to the stress ranges previously computed. The maximum strain levels in each test were held constant to simulate a residual tensile strain within the rail head, while the magnitudes of the negative strain excursions from that maximum strain were selected to achieve long life, but nonrunout fatigue crack initiation conditions.

Three different representations (histories) of the 1/2FEC + 1/2UP spectrum were tested. Only the simplest (unit train) history representation was used for the NEC and SP spectra. The unit train history was an 8-level high-to-low, block loading history (applied under strain control), representing a single train which was repeated continuously. It was designed to represent an average or unit train. A total of 170 unit trains (1 MGT) contained the same number of cycles at each strain level as the same number of trains of the more complex spectra.

The next history of intermediate complexity was the train-by-train loading pattern, which consisted of an 11-level series of high-to-low block loading (under strain control). This history was made up of six different trains, as shown in Figure 19. The composition of each of the six trains was selected more or less arbitrarily, however, they resembled actual trains in size and load content. A mixture of 2 trains A1, 6 trains A2, 12 trains A3, 120 trains B, 20 trains C, and 10 trains D were mixed in such a way that the heavy and light trains were not grouped together. The mixture of 170 trains was repeated during the tests.

The most complex history was random in nature with the 11 strain range levels within each train randomized and the six different train types within the history also randomized. An example of a random sequence of loads is shown in Figure 20.

A total of 17 variable amplitude strain control experiments were performed as listed in Table 8. Fatigue lives ranging from 127,000 to 6,480,000 cycles to failure were obtained. Data on stable stress levels for the various strain levels imposed in the unit train experiments are presented in Table 9.



FIGURE 19. TRAIN COMPOSITIONS FOR TRAIN-BY-TRAIN AND RANDOM HISTORIES



FIGURE 20. PART OF THE STRAIN HISTORY OF THE 11-LOAD RANDOM SERVICE SIMULATION EXPERIMENTS

TABLE 8. VARIABLE STRAIN AMPLITUDE FATIGUE TEST RESULTS

							<i>.</i> .					
Computed Cycles to Crack Initiation(a)		518,000 2,350,000	403,000 1,950,000	120,000 389,000	395,000 675,000	3,840,000(b) >6,480,000(c) 2,010,000		3,410,000 1,520,000(b)	· ·	378,000 713,000	4,110,000 2,090,000	
Cycles to Failure		551,000 2,500,000	2,070,000	127,570 413,360	420,110 717,650	4,080,000(b) 6,480,000(c) 2,140,000		3,630,000 1,620,000(b)		402,000 759,000	4,370,000 2,220,000	
Maximum Strain Range ∆ ⁶ Emax3		4.70 4.70	4.70	6.15 4.97	5.10 5.10	3.80 3.10 3.90		4.50 3.94	· · · · ·	5.49 5.49	4.50	
Stable Maximum Stress, ^o max, ksi	×	60.3 59.1	50.4	59.8 59.2	48.9 51.8	56.7 59.5 55.9		65.2 - (d)		58.4 59.6	59.4 63.3	
Maximum Strain, ^c max3 x 10 ⁻³		3.90 3.90	2.50		2.50 2.50	3.80 3.10 2.50		<b>3.20</b> 2.80		3.90 3.90	3.20 3.20	
Specimen Number		2-10-A 2-8-C 1-3-C	1-3-E	1-10-C 2-11-A	2-13-A 1-9-A	1-9-С 1-8-А 1-4-С		2-4-C 2-8-A		1-6-C 2-1-E	1-13-E 2-6-C	
Spectrum Description	<u>Unit Train</u>	∳FEC + ∳UP		NEC		SP	Train by Train	∔FEC + ₄UP	Random	4FEC + 4UP	· · ·	

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(a) Based on N_j = 94 percent of N_f. (b) Fillet failure.

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(c) Did not fail, runout.

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(d) Information not available.

TABLE 9. STABLE STRESS RESPONSE OF UNIT TRAIN TEST SPECIMENS

1

7.2 19.2 6.6 9.3 32.9 32.8 39.1 30.3 30.1 23.1 22.2 22.7 ω -10.3 5.3 15.3 15.3 15.3 8.8 4.5 4.5 -7.4 -2.1 -4.7 18.Ĵ 18.1 26.7 ~ -16.4 -11.6 -13.0 -8.4 -7.3 -7.8 -20.4 -4.5 4.1 3.4 14.4 -1.8 -1.6 -1.7 ø -7.0 -8.7 4.1 -30.7 -25.5 -21.6 -23.6 -14.3 -13.0 -13.6 -20.1 -17.5 -18.8 ഹ Minimum Stress, ksi Stress Level -32.9 -29.5 -31.2 -17.2 -21.3 -6.2 -37.9 -24.3 -23'.2 -23.6 -29.4 -27.3 -28.3 4 -25.5 -29.8 -14.2 -38.6 -37.0 -37.8 -30.8 -30.8 -30.8 -37.4 -34.6 -36.0 -43.0 ო -42.0 -40.1 -41.0 -49.1 -38.6 -45.2 -44.8 -45.0 -33.3 -38.0 -22.4 -36.4 -36.7 -36.5  $\sim$ -48.1 -48.9 -48.5 -54.2 -56.4 -55.3 -37.8 -43.1 -26.9 -55.0 -53.3 -54.1 -60.4 -51.9 56.7 55.9 59.5 60.3 59.1 59.7 59.8 59.2 48.9 51.8 50.4 52.4 50.4 51.4 Maximum Stress, σmax, ksi Åvg. Avg. Avg. Specimen Identification 2-10-A 2-8-C 1-10-C 2-11-A 2-13-A 1-9-A 1-9-C 1-4-C 1-8-A 1-3-C 1-3-E Spectrum Description FEC + FUP NEC SP ۰.

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#### 5. DAMAGE ACCUMULATION MODEL

The development of a damage accumulation model for crack initiation that would be useful in the reliability analysis of railroad rails involved several interrelated tasks, including 1) the definition of an initiation crack size, 2) the statistical definition of the constant amplitude crack initiation data on new and used rail material, and 3) the development of a damage model which could be used to predict variable amplitude crack initiation fatigue performance from constant amplitude crack initiation data. These tasks are reviewed in the following sections.

#### 5.1 DEFINITION OF INITIATION CRACK SIZE

It was shown earlier in Figure 9 that 85 to 95 percent of the total cycles to failure were involved in initiating a crack about 0.3-inch deep in the BCL specimen tested at R = -1.0. The work presented in this section was done to answer the following questions:(1) how is the transition crack size affected by stress ratio, and (2) at what initiation crack depth are stresses near the endurance limit sufficient to give crack tip stress intensities near the threshold for crack propagation? (Note that a crack initiated at a stress just above the endurance limit must be of sufficient size to be propagated at that same stress. Since the stress intensity depends on stress and crack size, the threshold stress intensity for crack growth can be exceeded only if the crack is large enough at a given stress.)

In reviewing past BCL work, (1) it is possible to determine that likely threshold stress intensity levels for rail steel at different stress ratios were as follows:

Stress Ratio	Stress Intensity Range at Threshold ∆K, ksi-in. ²
-1.0	14 - 18
0.10	12 - 16
0.50	. 7 - 9

In order to use this information effectively, it was necessary to know the relationship between crack depth and stress intensity for the test specimen being used in all fatigue tests. A recent publication by Dauod, et al, (16) shows that the effects of geometry on stress intensity are small (<3 percent) for crack depths less than 0.050-inch in the BCL and BCAC specimen of 0.250-inch diameter. In other words, the general formula for stress intensity could be used without a geometry factor as follows:

 $\Delta K = \Delta \sigma \sqrt{\pi a} , \text{ ksi-in.}^{1/2}$ (5)

Where  $\Delta \sigma$  = stress range, ksi

a = crack depth, inch

 $\pi = Pi (3.1416).$ 

Using the above expression, it was possible to develop constant stress-intensity-range lines on a plot of maximum stress versus crack depth. These curves are shown in Figures 21 through 23 for the test stress ratios of -1.0, 0.1 and 0.5. The approximate endurance limits obtained from constant amplitude fatigue experiments are also indicated. The intersection of the threshold stress-intensity lines with the endurance limit band creates a zone of particular interest. It is within this zone that stresses near the endurance limit also cause stress intensities near the threshold for crack propagation. The crack depths associated with this zone are logically associated with the transition crack size between crack initiation and propagation. The relationship between transition crack size and stress ratio is shown' in Figure 24. Obviously, the apparent equivalent initial flaw size decreases with increasing stress ratio. At an R = -1.0, a crack depth of 0.03-inch is representative, but crack depths as small as 0.006-inch are representative at an R = 0.5. The variable size of the transition flaw size shows its non-unique, empirical nature. It also illustrates



FIGURE 21. THRESHOLD VS. ENDURANCE LIMIT FOR R = -1.0



FIGURE 22. THRESHOLD VS. ENDURANCE LIMIT FOR R = 0.10







FIGURE 24. APPARENT OR EQUIVALENT INITIAL FLAW TRANSITION CRACK LENGTH (EIF) BETWEEN INITIATION AND PROPAGATION FOR RAIL STEEL FATIGUE SPECIMENS

the problem in the selection of a meaningful transition flaw size for a rail reliability analysis. A 0.03-inch flaw size was selected more or less arbitrarily, because crack propagation is expected at this crack size for all stress ratios above -1.0.

#### 5.2 STATISTICAL DEFINITION OF CONSTANT AMPLITUDE FATIGUE DATA

The variability in fatigue data is of prime importance in a rail reliability analysis. For this reason, all of the unused and used rail crack initiation data were reviewed to provide a statistical definition of those data.

#### 5.2.1 Unused Rail Crack Initiation Data

Through the use of the equivalent strain factor (Equation (1)), it was possible to combine all of the unused rail crack initiation data into a single fatigue curve, with the exception of those for transverse specimens. Grouped data of all the longitudinal orientations are shown in Figure 25. Some scatter is evident, but there is no consistent trend or layering of the data with either stress ratio or specimen location within the rail head. Two other things are also noted. First, the limited data for periodic overstrains shown also illustrate the dramatic effect of overstrains on high cycle fatigue performance. Second, the percent of runouts (shown for log  $\varepsilon_{eq}$ value of -2.55 and below) indicates that the rate of increase in percent of runouts depends upon the stress ratio. These data are not conclusive but perhaps should be taken into account in reliability analyses involving constant amplitude fatigue data near the so-called endurance limit. The periodic overstrain data seem to indicate that the endurance limit found in the constant amplitude tests may not apply for variable amplitude loading conditions.

In order to assess the variability of the data for different life ranges, the data were subdivided into four groups as follows:

Group	Equivalent Strain						
1	$\epsilon_{eq} < 4.40 \times 10^{-3}$						
2	$3.20 \times 10^{-3} \le \epsilon_{eq} < 3.60 \times 10^{-3}$						
3	$3.00 \times 10^{-3} \le \varepsilon_{eq} < 3.20 \times 10^{-3}$	, ' ,					
4	2.80 x $10^{-3} \le \epsilon_{eq} < 3.00 x 10^{-3}$	•,					

Most tests completed below an  $\varepsilon_{eq}$  of 2.80 x 10⁻³ were runouts and were therefore not included in this analysis.

The data within each group were ranked and the statistical parameters shown in Table 10 were calculated. These statistics include an average life 10 log  $N_i$ , a coefficient of variation, median ranks (for plotting on log-normal paper) and fraction failed (for plotting on Weibull paper). The average life, of course, goes up with decreasing equivalent strain. The coefficient of variation also increases somewhat with increasing life which has been noted elsewhere with some steel alloys. This increase in scatter is evident in Figure 26 where the four groups of data are displayed on log-normal probability paper. The increase in slope of the data trends with increasing crack initiation life indicates increased variability. Several observations can be made here. First, in the area of principal interest (early failures in the long life regime) the distribution of failures follows a log-normal trend reasonably well. Second, the breakpoint in the curve for the longest life test series seems to be indicative of a transition in the failure mechanism. It is at least plausible that the early failures preceding the transition are representative of specimens already containing microcracks, and the late failures (and runouts) following the transition are representative of essentially defect-free rail specimens. Whatever the explanation, the area of prime interest involving early failures follows log-normal trends.

The data were also plotted on two-parameter Weibull paper as shown in Figure 27. At the shorter lives (Groups 1 and 2), there are too few data to verify trends, but for Groups 3 and 4, the data fairly clearly do not follow a two-parameter Weibull, unless one resorts to a two or three piece linear representation of the data.





FIGURE 26. LOG NORMAL PROBABILITY PLOT OF UNUSED RAIL CRACK INITIATION DATA



FIGURE 27. TWO-PARAMETER WEIBULL OF UNUSED RAIL CRACK INITIATION DATA

Fraction Failed

TABLE 10. GROUPED AND RANKED CRACK INITIATION DATA ON UNUSED RAIL MATERIAL

GROUP	RANK	FATIGUE LIFE	AVERAGE LIFE (LOG)	COEFFICIENT OF VARIATION, %	MEDIAN RANK	FRACTION FAILED
1	1 2 3	10775 11770 14330	12203	1.5	0.2063 0.5000 0.7937	0.33 0.67 1.00
2	1 2 3 4 5	49710 57000 57610 62000 77690	60133	1.5	0.1294 0.3147 0.5000 0.6853 0.8706	0.2 0.4 0.6 0.8 1.0
3	1 2 3 4 5 6 7 8 9 10 11 12 13 14 15 16 17 18 19 20 21 22 23 24 25	51000 60000 65000 72000 79000 88000 91000 92580 102000 107000 107000 107000 15000 144000 163800 163800 163800 167000 177000 183300 187800 234700 246800 308000 924600	128653	5.3	0.0277 0.0670 0.1064 0.1457 0.1851 0.2245 0.2638 0.3032 0.3425 0.3819 0.4212 0.4606 0.5000 0.5393 0.5787 0.6180 0.6574 0.6967 0.7361 0.7754 0.8148 0.8542 0.8935 0.9329 0.9722	0.04 0.08 0.12 0.16 0.20 0.24 0.28 0.32 0.36 0.40 0.44 0.48 0.52 0.56 0.60 0.64 0.68 0.72 0.76 0.80 0.84 0.88 0.92 0.96 1.00
4	1 2 3 4 5 6 7 8 9 10 11 12 13 14 15 16 17 18 19 20 21 22 23	81000 102000 127870 143000 157000 167000 173940 179290 183000 209000 210000 254000 256000 294000 303000 309000 315000 328000 334000 358000 431000	317924	6.5	0.0198 0.0480 0.0763 0.1043 0.1327 0.1610 0.1892 0.2175 0.2457 0.2740 0.3022 0.3305 0.3587 0.3870 0.4152 0.4435 0.4152 0.4435 0.4717 0.5000 0.5282 0.5564 0.5847 0.6129 0.6412	0.029 0.057 0.086 0.114 0.143 0.171 0.200 0.229 0.257 0.286 0.314 0.343 0.371 0.400 0.429 0.457 0.486 0.514 0.543 0.571 0.600 0.629 0.657

TABLE 10. (Concluded)

GROUP	RANK	FATIGUE LIFE	AVERAGE LIFE (LOG)	COEFFICIENT OF variation, %	MEDIAN RANK	FRACTION FAILED
	24 25 26 27 28 29 30 31 32 33 33 34 35	438000 589000 610000 645000 712000 973000 1314000 1324000 3026000 5080000R 5087000R 5100000R			0.6694 0.6977 0.7259 0.7542 0.7824 0.8107 0.8389 0.8672 0.8954 0.9237 0.9219 0.9801	0.686 0.714 0.743 0.771 0.800 0.829 0.857 0.886 0.914 0.943 0.971 1.000

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In summary, it appears that a log-normal or a two-parameter Weibull representation of the early failure data is useful only if the late failure nonlinearities can be ignored. This may well be the most reasonable approach, especially in view of the lack of ready alternatives. An assumption of homoscedasticity is also questionable, if such an assumption is used, it should be based on the slope of the low equivalent strain data groups in the early failure region (see Figure 26, data groups 3 and 4 below about a 60 percent median rank).

#### 5.2.2 USED RAIL CRACK INITIATION DATA

Most of the used rail initiation data were generated at maximum stress levels of 85 and 95 ksi (R = 0.10). At the other three stress levels (of 105, 75, and 65 ksi), data on only one or two of the four used rail materials were generated — the fatigue lives obtained at these levels were, therefore, not representative of the overall collection of used rail materials and were omitted from this analysis.

The data generated at the two primary stress levels were ranked for statistical analysis in the manner shown in Table 11. The ranked data were plotted on normal probability paper in Figure 28 to evaluate their trends. Distinctly nonlinear patterns are evident at both stress levels. Since this result was considered unsatisfactory, the data were examined jointly, in an effort to better represent overall used rail fatigue data trends. Figure 29 shows the result of the data combination. A very nearly linear pattern is evident. The average fatigue life of this combined group was 147,900 with a coefficient of variation of 10.0.

The combined used rail fatigue data display mean fatigue life is nearly equal to the unused rail data (at log  $\varepsilon_{eq} = -2.53 \ \overline{N_i} = 147,900$  for used rail, 180,000 for unused rail). The coefficient of variation of 10.0 for the used rail data is nearly double that found for unused rail material, however, and should be taken into account in the reliability analysis.

#### 5.3 VARIABLE STRAIN AMPLITUDE CRACK INITIATION PREDICTION

Three different methods of predicting fatigue lives for the variable strain amplitude experiments were attempted. The basic element in each calculation was the equivalent strain factor, previously presented in Equation (1). Since strain was controlled in these spectrum tests, the various levels of strain range,  $\Delta \epsilon$ , were known and the maximum stable stress,  $\sigma_{max}$ , was necessarily measured or calculated. Where  $\sigma_{max}$  was computed, the cyclic stress-strain curve for the material was used.

Damage calculations for the three-unit train spectra are shown in Tables 12 through 14. Similar calculations for the train-by-train and random spectra are shown in Table 15. The individual methods of linear damage calculations involved the following variables:

Method	<u>Fatigue Curve</u>	Maximum Stress
. 1	Constant Amplitude	Actual
2	Periodic Overstrain	Actual
3	Periodic Overstrain	Computed

The results of these three damage calculations for all the spectra are summarized in Table 16. The results are presented graphically in Figures 30, 31, and 32 for Methods 1, 2, and 3, respectively. It is evident in Figure 30 that Method 1 overestimated actual crack initiation lives in nearly every case, in an unconservative manner. In Figure 31, it is evident that Method 2 provides a substantially improved prediction of actual crack initiation lives. For reference purposes, 1 and 2 standard deviation lines are constructed (based on a constant coefficient of variation of 5 percent). All data are contained within two standard deviations. TABLE 11. GROUPED AND RANKED CRACK INITIATION DATA ON USED RAIL MATERIAL

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FRACTION FAILED	0.048 0.095 0.143	0.190 0.238 0.286	0.333 0.429	0.524	0.667	0.762	0.071 0.143-0.214 0.286 0.357	0.429	0.571	0.714	0.857 0.929 1.00
MEDIAN RANK	0.033 0.080 0.126	0.173 0.220 0.226	0.360	0.500 0.500 0.547	0.593 0.640 0.687	0.780	0.048 0.118-0.0187 0.257 0.326	0.396	0.535	0.674	0.813 0.882 0.952
COEFFICIENT OF VARIATION. PERCENT	9'6	:					10.6				
AVERAGE LIFE	173,400						121,900				
Fatigue Life	45 46 53	62 99 33	124	180	359 379 495	1064 3170 5000R	23,29 64 69	97	199	267 278	3/5 502 689
RANK	- N E	4001	- യ ന c	212	13	16 17 18-21	2 4°3	2	. თი	212	12 13 14
MAXIMUM STRESS LEVEL, ESTIMATED STRAIN RANGE AND EQUIVALENT STRAIN	85 ksi	$\Delta c = 2.75 \times 10^{-3}$ $c_{eq} = 2.75 \times 10^{-3}$			,		95 ksi	$\Delta c = 0.308$ $c = 3.16 \times 10^{-3}$	eq		

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FIGURE 28. USED RAIL DATA, STRESS LEVELS COMBINED



FIGURE 29. USED RAIL DATA, STRESS LEVELS PLOTTED SEPARATELY

CALCULATIONS	
DAMAGE	
FATIGUE	
TRAIN	
UNIT	
Ð	
HALF	
ONE -	
PLUS	
FEC	
ONE-HALF	
12.	
TABLE	

METHOD 3 DAMAGE INCREMENT (c,d) D _i x 10 ⁻⁵	3.26 7.63 6.48 12.00 10.47 9.40 2.44 <u>0.87</u> 52.55 5.71×10 ⁵	1.59 3.66 3.14 5.80 4.98 4.63 1.20 0.43 25.43 N f=1.18×10 ⁶
METHOD 2 DAMAGE INCREMENT(c) D ₁ x 10 ⁻⁵	2.17 5.05 4.29 4.29 6.81 6.16 1.57 <u>34.53</u> <u>34.53</u> <u>8.6</u> 9x10 ⁵	t 1.43 3.33 2.77 5.22 4.07 1.06 0.38 22.69 Nf=1.32x10 ⁶
METHOD I DAMAGE INCREMENT(b) D _i x 10-5	2.21 4.81 3.26 3.54 1.16 0.29 0.29 0.29 0.29 0.29 0.29 0.29 0.29	t 1.50 2.38 1.32 1.30 0.10 0.10 0.10 6.98 6.98 f=4.30x10 ⁶
EQUIVALENT STRAIN Eeq (a) x 10-3	3.48 3.12 2.77 2.25 1.45 1.45	3.28 2.94 2.79 2.12 1.75 1.37
CYCLE PER BLOCKS, C ₁	200,00338655 -	
STABLE STRESS RANGE, Δσ, ski	108.2 96.2 96.5 83.3 73.3 73.3 29.6 29.6	105.5 92.4 87.4 79.7 70.2 59.2 28.7 28.7
STABLE MAXIMUM STRESS ^o max ksi	59.7	51.4
STRAIN RANGE,3 ∆∈ x 10 ⁻ 3	4.71 3.92 3.61 3.22 2.75 2.75 1.65 1.10	4. 71 3.92 3.61 3.22 2.75 1.10
MAXIMUM STRAIN,-3 ^E max x 10-3	3, 90	2.50
SPECIMEN NUMBER	2-10-A 2-8-C	1 - 3 - С 1 - 3 - Е

(a)  $\varepsilon_{eq} = (\Delta \varepsilon)^{"} (\sigma_{max}/E)^{1-m}$ , m = 0.60;  $E = 29.0 \times 10^{3}$  ksi.

(b)  $D_i = C_i/N_i$  where  $N_i = 2.249 \times 10^{-5} \epsilon_{eq}^{-3.76} + 4.365 \times 10^{-43} \epsilon_{eq}^{-18.76}$ . (c)  $D_i = C_i/N_i$  where  $N_i = 3.02 \times 10^{-13} \epsilon_{eq}^{-6.99}$ . (d)  $\sigma_{max}$  is computed from cyclic  $\sigma$ - $\epsilon$  curve.

TABLE 13. NEC UNIT TRAIN FATIGUE DAMAGE CALCULATIONS

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METHOD 3 DAMAGE(c,d) INCREMENT ¹ D _i x 10-5	7.20 24.97 22.74 43.07 42.96 41.93 20.93 11.95 2 <u>15.75</u> N_=1.39x10 ⁵	+ 3.93 6.62 6.62 12.53 12.17 12.17 3.51 5.34 3.51 66.34 - 4.52×10 ⁵	- 2.21 5.16 7.15 7.15 6.76 8.33 3.33 3.33 <u>37.48</u>
METHOD 2 DAMAGE(c) INCREMEN1 D _i × 10-5	13.21 15.41 11.27 21.22 21.22 20.39 20.39 10.26 5.80 1 <u>18.5</u> N_=2.53X10 ⁵	t 2.64 6.16 6.16 8.53 8.47 8.11 8.11 4.07 2.30 74.78 N_c=6.70x10 ⁵	- 3.92 3.22 6.12 6.12 5.79 7.79 1.67 33.92
METHOD 1 DAMAGE(b) INCREMENT D ₁ × 10-5	4.60 14.35 11.77 21.40 15.60 6.41 1.11 0.08 <u>75.32</u> N_=3.98x10 ⁵	t 2.58 3.55 3.55 4.19 2.02 0.02 0.01 0.01 0.01 0.01 0.01 0.01	- 3.76 1.83 1.83 0.87 0.25 0.25 0.04 
EQUIVALENT STRAIN, Eeq ^(a) x,10 ⁻³	4.09 3.66 3.19 2.67 2.42 2.02	3.58 3.21 2.99 2.80 2.34 1.77	3.41 3.05 2.85 2.23 2.23 1.69
CYCLES PER BLOCK,	1 5 1 1 1 2 0 6 0 1 2 0 1 2 0		
stable stress bunge, ÅG , ÅS Ì	120.2 108.9 97.7 97.7 97.7 97.7 70.1 52.6	111.7 98.4 90.8 84.2 76.1 76.1 76.1 76.1	105.7 95.4 88.2 81.6 53.4 55.1
MALE STRESS. ^O max > ks.i	59.8	59.2	50.4
STRAIN RANGE, ∆∈ x 10 ⁻³	6.15 5.11 4.54 4.07 3.03 3.03 3.03 1.90	4.97 3.67 3.267 2.29 1.54 1.54	5.10 3.77 3.377 2.59 2.59 1.58 1.58
MAXIMUM STRAIN, ^E max x 10 ⁻³	4.70	3. 80	2.50
SPECIMEN	1-10-C	2-11-A	2-13-A 1-9-A

(a)  $\varepsilon_{eq} = (\Delta \varepsilon)^m (\sigma_{max}/E)^{1-m}$ , m = 0.60;  $E = 29.0 \times 10^3 \text{ ksi}$ . (b)  $D_i = C_i/N_i$  where  $N_i = 2.249 \times 10^{-5} \varepsilon_{eq}^{-3.76} + 4.365 \times 10^{-43} \varepsilon_{eq}^{-5}$ . (c)  $D_i = C_i/N_i$  where  $N_i = 3.02 \times 10^{-13} \varepsilon_{eq}^{-6.99}$ .

-18.76.

(d)  $\sigma_{max}$  is computer from cyclic  $\sigma$ - $\varepsilon$  curve.

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TABLE 14. SP UNIT TRAIN FATIGUE DAMAGE CALCULATIONS	STRAINWILL START No.METHOD 1METHOD 2METHOD 3RANGE, $\Delta \varepsilon \times 10^{-3}$ $\sigma_{max}$ $\Delta_{\sigma}$ PER $\varepsilon_{eq}$ EQUIVALENTDAMAGE (b)(DAMAGE (c)DAMAGE (c,d) $\Delta \varepsilon \times 10^{-3}$ $v_{si}$ $v_{si}$ $v_{si}$ $v_{si}$ $v_{si}$ $v_{si}$ $v_{si}$ $v_{si}$ $\Delta \varepsilon \times 10^{-3}$ ksiksiBLOCK $\varepsilon_{eq}$ $v_{10^{-3}}$ $D_i$ $v_{10^{-5}}$ $D_i$ $v_{10^{-5}}$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	
LE 14. SP UNIT TRAIN	AME STRESS. NAME STRESS CYCLE AMEX. Add PER Cmax Ad PER ksi ksi BLOCK	56.7 94.5 1 90.0 5 82.2 6 73.9 18 63.7 30 52.6 60 38.6 60 23.8 120	59.5 86.4 1 73.7 6 65.7 18 65.7 18 55.4 30 45.1 60 20.4 120	55.9 99.0 1 93.9 5 85.7 6 85.7 18 77.2 18 64.6 30 52.5 60 37.8 60 23.1 120	
TAB	STRAIN RANGE, $\Delta \varepsilon \times 10^{-3}$	3.82 3.59 3.21 2.33 1.92 0.84	3.12 2.93 2.62 2.31 1.13 0.69	3.82 3.59 3.259 2.37 0.84 0.84	- U-
	MAXIMUM STRAIN, ^E max x 10 ⁻³	3.80	3.10	2.50	
	SPECIMEN	1-9-C	1-8-A	1-4-C	

(b)  $D_i = C_i/N_i$  where  $N_i = 2.249 \times 10^{-5} \epsilon_{eq}^{-3.76} + 4.365 \times 10^{-43} \epsilon_{eq}^{-18.76}$ . (c)  $D_i = C_i/N_i$  where  $N_i = 3.02 \times 10^{-13} \epsilon_{eq}^{-6.99}$ .

(d)  $\sigma_{\text{max}}$  is computed from cyclic  $\sigma\text{-}\epsilon$  curve.

SPECIMEN IDENTIFICATION	MAXIMUM STRAIN . E × 10 ⁻³	STABLE MAXI- MUM STRESS	STRAIN RANGE MULTIPLIER	STRAIN RANGE,	EQUIVALENT STRAIN, cX 10 ^{-3(b)}	CYCLE PER MGT, C.	DAMAGE INCREMENTS (c) D
-6-C and 2-1-E	3.9	68.9	1.407 1.320 1.320	Δε × 10 - 5.49 5.15	eq 4.04 3.88 3.38	- ~ ~ ~	0.10 × 10-3 0.35 × 10-3 0.35 × 10-3
			1.127	4.1	3.40 3.40	150 800	1.36 × 10_3 3.66 × 10_3 14.73 × 10_3
-			0.922 0.825 0.703	3.60 3.22 2.74	3.14 2.93 2.66	1000 3000 5000	10.56 × 10 3 19.52 × 10 3 16.55 × 10 3
			0.563 0.422 0.290	2.20 1.13	2.33 1.96 1.57	10,000 10,000 20,000	13.12 × 10 ⁻³ 3.92 × 10 ⁻³ 1.66 × 10 ⁻³
I-13-E, 2-6-C and 2-4-C	3.2	61.7	1.407 1.320	4.50	3.43	~ ~ œ	$\begin{array}{c} \Sigma 85.53 \times 10^{-3} \\ 2.0 \times 10^{-5} \\ 12.0 \times 10^{-5} \end{array}$
			1.223 1.127 1.055	3.38 3.38	3.15 3.01 2.89	40 150 800	43.2 × 10 ⁻⁵ 117.8 × 10 ⁻⁵ 472.9 × 10 ⁻⁵
		· · · · · ·	0.922 0.825 0.703	2.95 2.64 2.25	2.66 2.49 2.26	1000 3000 5000	331.1 × 10 ⁻⁵ 626.0 × 10 ⁻⁵ 529.9 × 10 ⁻⁵
			0.563 0.422 0.290	1.80 1.35 0.93	1.98 1.67 1.33	10,000 10,000 20,000	$\begin{array}{c} 420.4 \times 10^{-5} \\ 127.9 \times 10^{-5} \\ 52.1 \times 10^{-5} \end{array}$
2-8-A	2.8	57.1	1.407	3.94	3.07	- - -	· 22735.3 × 10 ⁻³
			1.320 1.223	3.70 3.42	2.96	4 ⁰ 8	5.59 × 10 ⁻⁵
			1.127 1.055	3.16 2.95	2.69 2.58	150 800	$53.71 \times 10^{-5}$
			0.922 0.825	2.58	2.38	1000	152.15 × 10 ⁻⁵
			0.703	1.97	2.03	5000	250.24 × 10-5
			0.422	8.1 1.18	1.78	10,000	$199.72 \times 10^{-5}$
		. v	0.290	0.81	1.19	20,000	23.94 x 10 ⁻⁵ Σ1268.22 x 10 ⁻⁵
(a) $\sigma_{max} = 1.758$	x 10 ³ _{Ea} 0.583 for _E	a ^{- c} max - 1.344	x 10 ⁻³ .				
$(\mathbf{p}) \; \varepsilon_{2} = (\Delta \varepsilon)^{m} \; (\varepsilon)$	J/E) ^{l_m} where m	= 0.60, E - 27.	0 × 10 ⁶ .			•	,
ha .	max						

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(c)  $D_{i} = C_{i}/N_{i}$  where  $C_{i} = 3.090 \times 10^{-4} \varepsilon_{eq}^{-3.265}$  for  $\varepsilon_{eq}^{-3.802} \times 10^{-3}$ .

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TABLE 15. FATIGUE DAMAGE CALCULATIONS FOR RANDOM AND TRAIN-BY-TRAIN SPECTRA (METHOD 3)

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TESTS	N _i , ACTUAL ^(a)	$\begin{array}{c} 5.18 \times 10^{5} \\ 2.35 \times 10^{5} \\ 2.35 \times 10^{5} \\ 4.03 \times 10^{6} \\ 1.95 \times 10^{6} \end{array}$	$\begin{array}{c} 1.20 \times 10^{5} \\ 3.88 \times 10^{5} \\ 3.95 \times 10^{5} \\ 3.95 \times 10^{5} \\ 6.75 \times 10^{5} \end{array}$	>6.48 × 10 ⁶ (b) 3.84 × 10 ⁶ 2.01 × 10 ⁶	3.78 × 10 ⁵ 7.13 × 10 ⁵ 7.13 × 10 ⁶ 4.11 × 10 ⁶ 2.09 × 10 ⁶	3.41 × 10 ⁶ 1.52 × 10 ⁶
IABLE AMPLITUDE	N _i , BASED ON PERIODIC OVERSTRAIN CURVE AND CALCULATED, ^G max	5.71 × 10 ⁵ 5.71 × 10 ⁵ 5.71 × 10 ⁶ 1.18 × 10 ⁶ 1.18 × 10 ⁶	$\begin{array}{c} 1.39 \times 10^{5} \\ 4.52 \times 10^{5} \\ 8.00 \times 10^{5} \\ 8.00 \times 10^{5} \end{array}$	3.55 × 10 ⁶ 1.08 × 10 ⁶ 2.13 × 10 ⁶	5.85 × 10 ⁵ 5.85 × 10 ⁵ 5.85 × 10 ⁶ 1.83 × 10 ⁶ 1.83 × 10 ⁶	1.83 x 10 ⁶ 3.94 x 10 ⁶
TO FAILURE FOR VAR1	N _i , BASED ON PERIODIC OVERSTRAINS CURVE AND ACTUAL, o _{max}	8.69 × 10 ⁵ 8.69 × 10 ⁵ 8.69 × 10 ⁶ 1.32 × 10 ⁶ 1.32 × 10 ⁶	$\begin{array}{c} 2.53 \times 10^{5} \\ 6.70 \times 10^{5} \\ 8.84 \times 10^{5} \\ 8.84 \times 10^{5} \\ 8.84 \times 10^{5} \end{array}$	3.71 × 10 ⁶ 1.80 × 10 ⁶ 1.87 × 10 ⁶		
ESTIMATED AND ACTUAL SPECIMEN CYCLES	N ₁ , BASED ON CONSTANT AMPLITUDE CURVE AND ACTAUL, o _{max}	1.96 × 10 ⁶ 1.96 × 10 ⁶ 4.30 × 10 ⁶ 4.30 × 10 ⁶	$\begin{array}{c} 3.98 \times 10^{5} \\ 1.55 \times 10^{6} \\ 2.82 \times 10^{6} \\ 2.82 \times 10^{6} \\ 2.82 \times 10^{6} \end{array}$	$\begin{array}{c} 3.97 \times 10^{7} \\ 7.67 \times 10^{6} \\ 7.74 \times 10^{6} \end{array}$		
	SPECIMEN NUMBER	2-10-A 2-8-C 1-3-C 1-3-E	1-10-C 2-11-A 2-13-A 1-9-A	1-8-A 1-9-C 1-4-C	1-6-C 2-1-E 1-13-E 2-6-C	2-4-C 2-8-A
TABLE 16.	SPECTRUM DESCRIPTION	Unit Train ≜FEC + §UP	Unit Train NEC	Unit Train SP	Random 불FEC+ 훨UP	Train-by-Train ŁFEC + ŁUP

(a) Computed as  $94\% \cdot of$  actual N_f.

(b) Did not fail.

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FIGURE 30. METHOD 1, CONSTANT AMPLITUDE CURVE AND ACTUAL MAXIMUM STRESSES (UNIT TRAIN ONLY)



FIGURE 31. METHOD 2, PERIODIC OVERSTRAIN WITH ACTUAL  $\sigma_{max}$ 



FIGURE 32. METHOD 5, PERIODIC OVERSTRAIN WITH CALCULATED Tmax

The predictions based on Method 3 are shown in Figure 32. A similar predictive capability is evident for Method 3, as was shown for Method 2, although it does tend to be somewhat more conservative in the long-life regime. Both Methods 2 and 3 provide predictions nearly as good as the basic scatter in the data.

Neither of the two satisfactory methods included any history or sequence accountability; they represented simple, Miner's rule, linear damage models based on strainlife fatigue curves for test with overstrain cycles. Interestingly, it does not appear that history effects were particularly significant. It is also evident, however, that the second and third damage models were somewhat conservative in the longlife tests which suggests that smaller cycles in these histories may have been nondamaging early in the history, thereby prolonging life in the manner discussed previously for the periodic overstrain data. Lacking a simple and justifiable method for introducing a history dependence in the damage models investigated, it is recommended that the third linear damage method be used in a reliability analysis. The third method is suggested over the second simply because actual stable maximum stresses in a history are seldom known. It should be noted, however, that its adequacy has been proved only for the life ranges covered by the tests. Therefore, generalization is still speculative.

#### 5.4 IMPLEMENTATION OF DAMAGE MODEL IN A RELIABILITY ANALYSIS

The following procedure should be used to perform life estimates for a stresscycle history.

- Sum all cycles at like values of local stress range and maximum stress within the spectrum.
- 2. Compute a strain range for each stress range using the following definition of the cyclic stress-strain curve

$$\varepsilon_{a} = \frac{S_{a}}{29 \times 10^{3}}, S_{a} \le 39.1$$

$$\varepsilon_{a} = \left(\frac{S_{a}}{3.43 \times 10^{3}}\right), 1.48$$

$$39.1 \le S_{a} \le 51.3$$

$$\varepsilon_{a} = \left(\frac{S_{a}}{8.05 \times 10^{2}}\right)^{2.26}, S_{a} \ge 51.3$$

- 3. Using the computed values of strain range and known values of maximum stress, compute equivalent strains. If maximum stress is negative, change the sign of minimum and maximum stresses before computing equivalent strains.
- Using the number of cycles of each stress magnitude, compute damage increments using Method 3, i.e., strain-life curve corrected for effect of periodic overstrain.
- 5. Sum the damage increments for the spectrum.
- 6. Consider the inverse of the total damage increment as the predicted average number of spectra to fatigue crack initiation.

From this experimental and analytical investigation of the crack initiation behavior of rail steels, the following conclusions have been made:

- 1. Periodic overstrains above the constant amplitude endurance limit will substantially increase the damage caused by strain ranges below that limit, and a periodic overstrain fatigue curve should be used in life predictions on such spectra.
- 2. Transverse rail head crack initiation properties can be expected to fall well below longitudinal rail head properties.
- 3. Linear damage accumulation models can be expected to provide reasonable life predictions, although such life estimates may become conservative in the very long life histories.
- 4. There is no definitive transition flaw size, but a flaw of 0.030-inch may serve as a useful transition flaw between crack initiation and propagation.

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